# **Inorganic Chemistry**

# Structural Assembly from Phosphate to Germanophosphate by Applying Germanate as a Binder

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**S** Supporting Information

[AB](#page-2-0)STRACT: [Structural](#page-2-0) [as](#page-2-0)sembly from phosphate to germanophosphate by applying germanate as a binder has been achieved. Two isotypic porous compounds,  $\text{K}_3[\text{M}^{\text{II}}_4(\text{HPO}_4)_2][\text{Ge}_2\text{O}(\text{OH})(\text{PO}_4)_4]\cdot x\text{H}_2\text{O}$   $(\text{M}^{\text{II}} = \text{Fe},$ Cd;  $x = 2$  for Fe and 3 for Cd, denoted as  $KFeGePO-1$ and KCdGePO-1, respectively), contain a known transition-metal phosphate (TMPO) layer,  $\frac{2}{\infty}$ {[M<sub>2</sub>(HPO<sub>4</sub>)<sub>3</sub>]<sup>2-</sup>}, which is built from chains of transedge-sharing  $MO<sub>6</sub>$  octahedra bridged by  $MO<sub>5</sub>$  trigonal bipyramids that were further linked and decorated by phosphate tetrahedra. The layers are bound by infinite chains of  $GeO<sub>s</sub>(OH)$  octahedra, resulting in a 3D openframework structure with 1D 12-ring channels that are occupied by K<sup>+</sup> ions and water molecules. The curvature of the TMPO layers and shape of the 12-ring windows can be tuned by the transition metals because of their Jahn− Teller effect.

Germanates have been approved to have extra-large pores in<br>their structure and may have promising application as candidates to replace silicates in gas absorbents, catalysts, and separators in the chemical and petroleum industries.<sup>1</sup> Incorporating heteroatoms into germanate frameworks increases the structural diversity and generates another class of ma[te](#page-2-0)rials with interesting structures and properties.<sup>2</sup> Transition metals and rare earths, containing a d- or f-shell electron, are candidates to obtain compounds with interesting physica[l](#page-2-0) properties, such as optical, magnetic, and electric properties.<sup>3</sup> However, it is a great challenge to directly obtain transition-metal germanates without applying organic ligands or co[ord](#page-2-0)inated groups. To our knowledge, phosphate has a strong affinity to transition metals.<sup>4</sup> On the basis of this feature, one can expect to fuse transition metals to germanates by adding phosphate, which may be able t[o](#page-2-0) overcome the obstacle and become a heteroanionic system for gaining considerable structural diversity and useful physical properties.

The resulting heteroanionic system is germanophosphate (GePO), which is a quite young subject with regard to germanates. Many features of GePOs are still waiting to be discovered. To date, most of the known GePOs have been prepared by conventional solid-state reactions at high temperatures, which lead to dense structures.<sup>5</sup> By use of a hydro/ solvothermal method, several transition-metal GePOs have been synthesize[d](#page-2-0).<sup>6</sup> Very recently, we obtained a series of transitionmetal GePO compounds, $^7$  whose structures consist of zigzag metal octa[h](#page-2-0)edral chains linked by germanium phosphate pentamers. However, th[e](#page-2-0)se previously reported transitionmetal GePOs, except the zinc compound,<sup>6a</sup> did not exhibit expected porous features. During our investigations on porous GePOs, by applying alkali metal as the templ[ate](#page-2-0), we successfully obtained two isotypic porous compounds, KFeGePO-1 and KCdGePO-1, containing 12-ring channels. In the structure, a known transition-metal phosphate layer was bound by germanate chains to a porous framework structure.

KFeGePO-1 and KCdGePO-1 have been solvothermally synthesized. For KFeGePO-1, a mixture with a molar ratio of  $GeO_2/Fe/KCl/H_3PO_4/H_2O/1,2$ -propanediol/triethylamine = 0.7:0.25:2:14.6:111:21:14.4 was transferred into a Teflon-lined stainless-steel autoclave (15 mL in volume) and heated at 190 °C for 5 days under static conditions. A similar process was applied for KCdGePO-1, but  $Cd(AC)_2$ ·2H<sub>2</sub>O (0.266 g, 1 mmol) takes the place of iron powder. Colorless prismatic crystals in high yield (about 80%, based on Fe/Cd) were obtained. The phase purities of two compounds have been checked by performing powder X-ray diffraction (PXRD). Qualitative energy-dispersive X-ray analysis results confirmed the presence of K, Ge, Fe/Cd, and P (Figure S1 in the Supporting Information, SI). IR spectra confirmed the presence of a hydroxyl (OH) group and water molecules (Figure S2 i[n the SI\). Thermal ana](#page-2-0)lysis results of KFeGePO-1 showed a two-step weight loss with a total amount of 5.6%, which agrees well [wit](#page-2-0)h the removal of two water molecules and the condensation of three OH groups (calcd 5.6%; Figure S3a in the SI). For KCdGePO-1, a similar thermal decomposition process has been taken: because of its one additional channel [wa](#page-2-0)ter, the total weight loss is about 5.7% (calcd 5.9%, for the removal of three water molecules and three OH groups). The divalent state of  $Fe^{2+}$  was confirmed by

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<span id="page-1-0"></span>Mössbauer spectrometry (Figure S4 in the SI). Magnetic susceptibility results reveal that KFeGePO-1 exhibits antiferromagnetic properties at lower temperatures ([Fig](#page-2-0)ure S5 in the SI).

 $KFeGePO-1$  and  $KCdGePO-1<sup>9</sup>$  (Figure 1) have similar [fra](#page-2-0)mework structures but differ in the channel shape and channel



Figure 1. Framework conformation of KFeGePO-1: (a) Two differently oriented ferrous ribbons, built from chains of trans-edgesharing FeO<sub>6</sub> octahedra bridged by FeO<sub>5</sub> trigonal bipyramids, stacked along [010] in an "ABAB" fashion. (b) The successive ribbons are stapled by  $HPO_4$  tetrahedra and decorated by  $PO_4$  groups on either side to form a layer. (c) A  $\text{GeO}_5(\text{OH})$  octahedral single chain is shown. (d) The layers are bound by germanate chains to result in a 3D framework with 1D 12-ring channels running along  $[001]$  in which  $K^+$  ions and water molecules reside. Color code:  $FeO<sub>6</sub>/FeO<sub>5</sub>$  polyhedra, blue;  $GeO<sub>5</sub>(OH)$  octahedra, green; PO<sub>4</sub> tetrahedra, orange; K atoms, medium-gray balls; O atoms, red balls; H atoms, dark-gray balls.

species arrangement. Here we choose the iron compound as an example to illustrate the conformation of their frameworks. In **KFeGePO-1**, the crystal structure is built from  $FeO<sub>6</sub>$  octahedra, FeO<sub>5</sub> trigonal bipyramids, GeO<sub>5</sub>(OH) octahedra, and HPO<sub>4</sub> and

PO4 tetrahedra. The framework contains two orientationdifferent ferrous ribbons, built from chains of trans-edge-sharing FeO<sub>6</sub> octahedra running along [001] loop-branched by FeO<sub>5</sub> trigonal bipyramids in an up-and-down manner alternately, which are stacked along [010] in an "ABAB" fashion (Figure 1a). The resulting ribbons are further interconnected by  $HPO<sub>4</sub>$ groups to form an undulated layer parallel to the bc plane, with additional  $PO_4$  tetrahedra decorating either side (Figure 1b). Such ferrous phosphate layers are linked by infinite chains of trans-corner-sharing  $GeO<sub>5</sub>(OH)$  octahedra, resulting in a 3D framework structure with 1D 12-ring channels parallel to the c axis in which  $K^+$  ions and water molecules reside (Figure 1c). The inorganic framework can maintain after removal of the channel water, as indicated by in situ PXRD (Figure S3b in the SI).

It is worth noting that the transition-metal phosphate (TMPO) layer has been found in a layered manganese p[hos](#page-2-0)phate  $(C_2H_{10}N_2)[Mn_2(HPO_4)_3(H_2O)]$  (see Figure 2a).<sup>8</sup> There the MnPO layer has the same conformation as that of the title compounds; the interlayer spaces are filled with et[hy](#page-2-0)lenediaminium ions and water molecules, which interlink the MnPO layers via hydrogen bonds. For the title compounds, the successive TMPO layers are clutched tightly by  $\text{GeO}_5(\text{OH})$  octahedral chains via common oxygen vertices, with four neighboring PO<sub>4</sub> groups leading to a relatively flexible 3D framework. That is to say, the 3D framework can be treated as a structural assembly from 2D TMPO layers by applying  $GeO<sub>5</sub>(OH)$  octahedral chains as binders (see Figure 2).

As shown in Figure 2, flat CdPO layers are observed in KCdGePO-1, whereas undulating FePO layers are found in KFeGePO-1. Because of the different curvature of TMPO layers, two almost equal pseudohexagonally shaped 12-ring channels



Figure 2. Top: Schematic presentation for the structural assembly from a 2D layer to a 3D framework by applying GeO6 single chains as binders. Middle: (a) layered structure of  $(C_2H_{10}N_2)[Mn_2(HPO_4)_3(H_2O)]$ ,<sup>8</sup> (b) flat layers bound by GeO<sub>6</sub> chains in KCdGePO-1, and (c) undulated layers bound by  $GeO<sub>6</sub>$  chains in KFeGePO-1. Bottom: topological presentation of KCdGePO-1 (b) and KFeGePO-1 (c) clearly showing the different shapes of the 12ring channels (outlined with blue and red lines). (d) Exper[im](#page-2-0)ental (black) and calculated (red) PXRD patterns of KCdGePO-1 and KFeGePO-1, with Cu Ka radiation. Color code: MnO<sub>x</sub> polyhedra, purple; CdO<sub>x</sub> polyhedra, light gray; FeO<sub>x</sub> polyhedra, blue; GeO<sub>5</sub>(OH) octahedra, green; PO<sub>4</sub> tetrahedra, orange; K atoms, medium-gray balls; O atoms, red balls; H atoms, dark-gray balls; C atoms, light-gray balls; N atoms, blue balls; Fe atoms, yellow-green balls; Cd atoms, light-gray balls; Ge atoms, turquoise balls; P atoms, purple balls.

<span id="page-2-0"></span>with aperture sizes of  $9.31 \times 7.38$  and  $9.76 \times 7.60$  Å were formed in the cadmium compound, while an elongated channel (10.48  $\times$ 5.65 Å) and a regular hexagonally shaped channel  $(8.20 \times 8.12)$ Å) are found for KFeGePO-1. Because of distortion of the channels, the Ow1 atom in the cadmium compound is assigned as full occupancy, while half-occupancy occurs in the iron compound.

The structure difference between KCdGePO-1 and KFeGe-PO-1 can be easily detected from their PXRD patterns. KFeGePO-1 has more reflections than KCdGePO-1, although both of them crystallize in the same orthorhombic space group, Cmcm (see Figure 2d). The difference between them may lie in the undulated degree of the TMPO layer, or even may be due to the Jahn−Teller eff[e](#page-1-0)ct of transition metals, as well as the channel species arrangement constrained by the channel shape. According to crystal-field theory,  $Cd^{2+}$  is a  $d^{10}$  ion, its electron density in the  $T_{2g}$  and  $E_g$  orbitals has  $O_h$  symmetry, and thus it has no Janh–Teller effect. However, for high-spin Fe<sup>2+</sup> (d<sup>6</sup>), the electronic state is not spatially symmetric; thus,  $E_g$  will spatially degenerate to lower the energy.<sup>10</sup> As a result, the bond distances of Fe−O vary in a larger range than those of Cd−O [magnitude of bond distortion  $(d_{\text{max}}/d_{\text{min}})$ : 1.11 and 1.03 for Fe1O<sub>6</sub> and  $Cd1O<sub>6</sub> octahedron, respectively; 1.18 and 1.11 for Fe2O<sub>5</sub> and$  $Cd2O<sub>5</sub>$  trigonal bipyramid, respectively]. The maximum torsion angle of Fe1−O6−Fe1 [104.0(2)°] is much larger than that of Cd1−O6−Cd1 [97.10(18)°] as well. The larger torsion angle of the iron compound makes its TMPO layers more undulating than that of the cadmium compound. It is worth noting that the different curvature of the TMPO layers also gives a hint of the structural flexibility.

In summary, the structural assembly from phosphate to GePO by applying germanate as a binder has been achieved. It is the first time to observe that a known 2D transition-metal phosphate layer {layers in  $(C_2H_{10}N_2)[Mn_2(HPO_4)_3(H_2O)]$ } can be bound by germanate units to form a 3D framework structure. The Jahn−Teller effect makes the iron compound have undulated layers while the cadmium compound has flat layers, which further results in different shapes of 12-ring channels and quite different PXRD patterns. Furthermore, the different curvatures of TMPO layers and channel shapes also give a hint of the structural flexibility. We suppose that Cu ions, having the strongest Jahn− Teller effect, may be able to be introduced into the structure. These kinds of structural assembly strategies may open a new vista for the design of new GePOs with diverse structures and interesting properties.

### ■ ASSOCIATED CONTENT

#### **S** Supporting Information

SEM images, FTIR spectra, TGA curves, in situ PXRD, Mössbauer spectra, magnetic properties, table of selected bond distances and angles, and crystallographic information files in CIF format. This material is available free of charge via the Internet at http://pubs.acs.org.

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#### Notes

The aut[hors declare no competin](mailto:yaxihuang@xmu.edu.cn)g financial interest.

## ■ ACKNOWLEDGMENTS

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(9) The crystal structures were solved and refined by using the SHELX programs ( Sheldrick, G. M. Acta Crystallogr. 2008, A64, 112−122 ). Crystal data: orthorhombic, Cmcm (No. 63), at 173 K. KFeGePO-1:  $a =$ 18.672(6) Å,  $b = 17.990(6)$  Å,  $c = 6.718(2)$  Å,  $V = 2256.6(12)$  Å<sup>3</sup>,  $Z = 4$ ;  $D_{\rm calc} = 3.316 \text{ g} \cdot \text{cm}^{-3}$ ; 130 free parameters; GOF = 1.158; R1 = 0.0486 [I  $> 2\sigma(I)$ ], wR2 = 0.0996 for all data; residual electron density (max/min)  $= 1.203/-1.133$  e·Å<sup>-3</sup>. KCdGePO-1: a = 19.069(5) Å, b = 19.104(5) Å,  $c = 6.7711(17)$  Å,  $V = 2466.7(11)$  Å<sup>3</sup>,  $Z = 4$ ;  $D_{calc} = 3.692$  g·cm<sup>-3</sup>, 126 free parameters; GOF = 1.078, R1 = 0.0377  $[I > 2\sigma(I)]$ , wR2 = 0.0994 for all data; residual electron density (max/min) =  $2.713/-1.037$  e·Å<sup>-3</sup>. . Detailed information may be found on the ICSD (see CSD-426507 and CSD-426506 for KFeGePO-1 and KCdGePO-1, respectively).

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